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# REPORT ON U<sub>3</sub>Si<sub>2</sub>-Al DISPERSION FUEL FOR HIGH-POWER RESEARCH REACTORS

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## ABSTRACT

The primary focus points of a report entitled “Technical Basis Report on Low Enriched Uranium Silicide-Aluminum Dispersion Fuel for High Power Research Reactors” are described in this paper. In the US Reactor Conversion program, U<sub>3</sub>Si<sub>2</sub>-Al dispersion fuel was developed. In 1988, the US NRC’s safety evaluation report (NUREG-1313) concluded that U<sub>3</sub>Si<sub>2</sub>-Al plate-type fuel was acceptable for use in research and test reactors with uranium densities up to 4.8 gU/cm<sup>3</sup> in the meat at the power level investigated in the demonstration tests. Fuel performance at the power conditions included in NUREG-1313 are summarized. As interest has recently been revived for higher power applications of this fuel, the effect of high-power conditions on fuel performance is addressed. In addition, the effect of U-densities higher than 4.8 gU/cm<sup>3</sup> is discussed. Knowledge gap items associated with higher power and higher U-density applications are identified together with the proposed resolution methods.

## 1. INTRODUCTION

A report on U<sub>3</sub>Si<sub>2</sub>-Al dispersion fuel in a plate form has been drafted as this fuel is undergoing renewed interest in potential applications for high power conditions, extending those evaluated in NUREG-1313 [1]. This fuel was licensed for use with uranium densities up to 4.8 gU/cm<sup>3</sup>, at the power level interrogated in the demonstration test (peak heat flux ≤ ~140 W/cm<sup>2</sup> and peak fuel temperature ≤ ~130 °C). Since the publication of NUREG-1313, more test data for this fuel, particularly at high powers, have become available.

The main objective of this report is to produce a stand-alone data compilation that enables review and analysis of the available higher-power data, extended from the data used for the qualification of this fuel in 1988. It is also indirectly intended to serve as a supporting document of the technical basis for future reactor-specific fuel qualification reports, in support of regulatory licensing activities for reactor conversions.

Knowledge gaps for higher-power applications of the fuel have been identified and indicate the desired work areas in the near future. Methods to resolve the information gaps are proposed.

The structure of this report is as follows:

Chapter 1: Background and basic information for the report are given.

Chapter 2: Fuel fabrication methods are described.

Chapter 3: Characterization of the as-fabricated fuel system is described.

Chapter 4: Fuel performance is discussed.

Chapter 5: Data and correlations of the properties of the fuel system are discussed.

Chapter 6: Fuel performance during off-normal states is addressed.

Chapter 7: Summary

Appendix: Gap analysis table

## 2. $U_3Si_2$ -Al PERFORMANCE

### 2.1 Lower Power Conditions

The crystalline U-Si fuel compounds ( $U_3Si$ ,  $U_3Si_2$  and USi) that form during fabrication, transform to non-equilibrium mixtures of amorphous  $U_3Si$ ,  $U_3Si_2$  and USi under irradiation at relatively low research reactor temperature conditions. When irradiated fuel is discussed, the amorphous states of those mixtures are of particular importance. A fuel particle can be a mixture of amorphous  $U_3Si$ ,  $U_3Si_2$ , and USi phases, each of which exhibits different fission gas bubble behavior. Therefore, the size of fission gas bubbles (FGB) can be drastically different locally depending on the local chemistry.

Low power conditions discussed in this section are for those interrogated in the past demonstration tests included in NUREG-1313 (having a heat flux  $\leq \sim 140$  W/cm<sup>2</sup> and a peak fuel temperature  $\leq \sim 130$  °C) [1],[2].

U-Si fuel meat swelling under low power conditions is primarily due to fuel particle swelling, since the effect of interaction layer (IL) growth between the fuel particles and the aluminum matrix is small. FGB swelling is the primary contributor to fuel particle swelling at high burnup. The kinetics of FGB growth is known to be dependent on the Si content in the fuel particle. The size of the FGB in  $U_3Si_2$  is much smaller than that in  $U_3Si$  (See Figure 1).

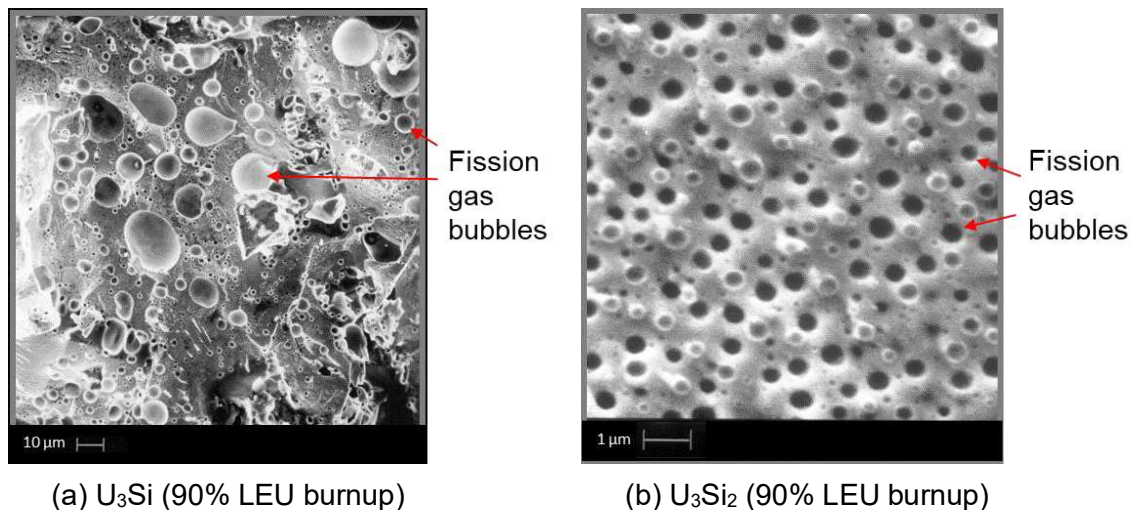


Figure 1: SEM images of the fractured surfaces of U-Si fuels irradiated at  $\sim 100$  °C in ORR. Note the difference in magnification [2].

In a lower magnification optical microscopy (OM) image (Figure 2), the microstructure of  $U_3Si_2$ -Al exhibits large FGB that are in-homogeneously distributed throughout the fuel meat. The non-uniform distribution of the large FGB implies that they are related to the local fuel chemistry. Nominally,  $U_3Si_2$  alloy contains a fraction of lower-Si content alloys such as  $U_3Si$ , which shows higher FGB swelling.

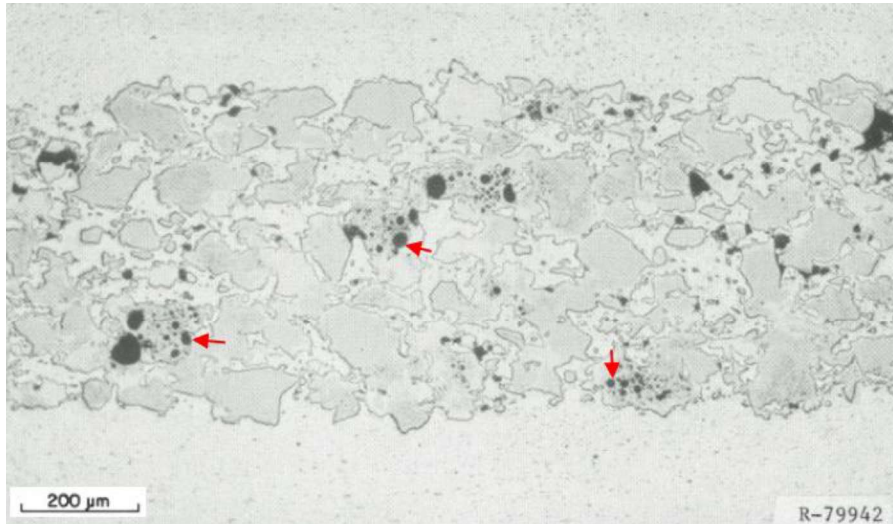


Figure 2: OM image of  $U_3Si_2$ -Al in BSI-201 at 71% LEU burnup. The red arrows mark large FGB [2].

A study at SCK•CEN (Belgium) showed that the size of the FGB is dependent on the Si content of the fuel particles they occupy, although the fuel is a nominal  $U_3Si_2$ -Al dispersion (Figure 3) [3]. Figure 3 shows that minor fuel phases such as  $USi$  and  $U_3Si$  are also found in  $U_3Si_2$ -Al dispersion fuel. The FGB are round in  $USi$  and  $U_3Si_2$ , but they take various shapes in  $U_3Si$ , implying signs of FGB coalescence.

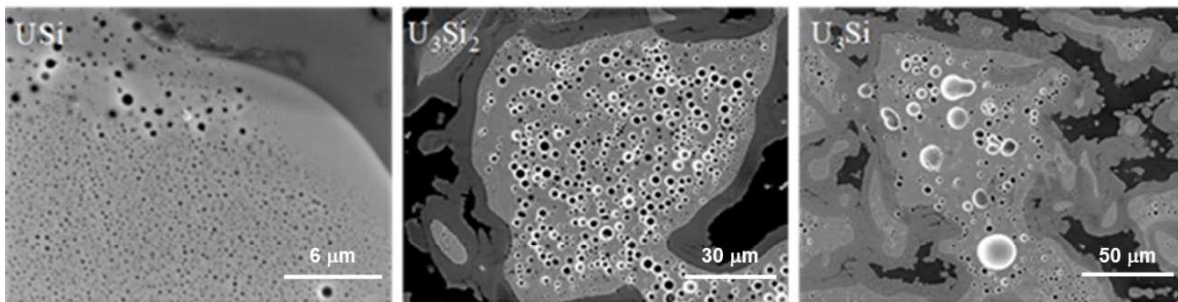


Figure 3: SEM images of fuel particles showing the differences in FGB generated in the  $USi$ ,  $U_3Si_2$  and  $U_3Si$  phases contained in a nominal  $U_3Si_2$ -Al dispersion fuel irradiated to a burnup of ~86% at ~130 °C. Note the difference in magnification.

The effect of Si content in U-Si fuel on overall fuel particle swelling is evident in Figure 4. The alloys with the least Si (indicated by the blue box in the figure legend) generally swell at a higher rate. This enhanced swelling is associated with a pronounced coarsening and coalescence of the fission gas bubbles (see the image of  $U_3Si$  in Figure 3). The data for higher Si content fuels such as  $U_3Si_2$  and  $USi$  do not deviate much from the dashed line for swelling due to solid fission products (the lowest possible fuel swelling) up to 100% LEU burnup, indicating no sign of breakaway swelling. The volume fraction of heterogeneously large FGB is still small, and more importantly the FGB are contained within the fuel particles as is shown in Figure 2 and Figure 3( $U_3Si$ ).

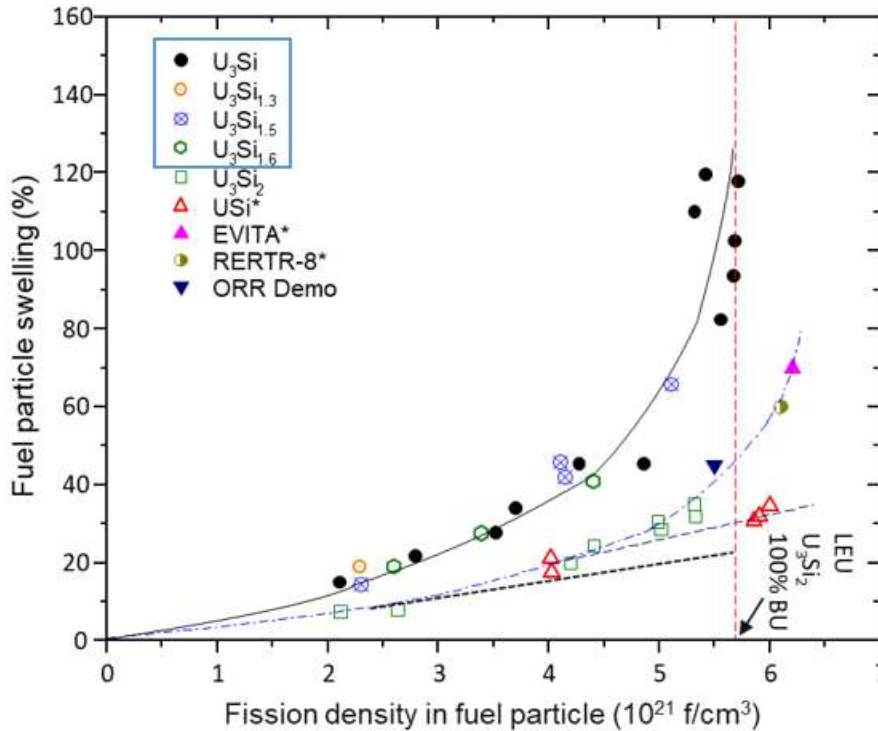


Figure 4: Fuel particle swelling in various U-Si/Al dispersion fuel plates. The Si/U ratios are nominal values. (\* Enriched in U-235 higher than 20%). EVITA [4], RERTR-8 [5] and ORR Demo [6] are  $U_3Si_2$ . The data for EVITA and ORR Demo are from full size plates; all other listed data are from reduced size plates. The dashed black line indicates swelling from only solid fission products.

## 2.2 High Power Conditions

High power conditions discussed here refer to irradiation studies with higher fission rates and fuel temperatures, but the same level of burnup as those discussed in Sect. 2.1.

Higher fission rates and temperatures do not appear to change the kinetics of FGB swelling significantly [2]. However, the IL growth rate is enhanced by higher fission rates and particularly higher temperatures. Under these conditions, the IL growth consumes the Al matrix more rapidly and, in some extreme cases, a complete depletion of matrix Al occurs. Continued irradiation under these conditions causes the IL density to increase (due to continued diffusion of the Al from the IL to the fuel), and pores to develop at the previous fuel particle surfaces (Figure 5). The formation of this pore type can lead to pillowing due to the fuel overheating as a result of the decreased thermal conductivity in the fuel meat.

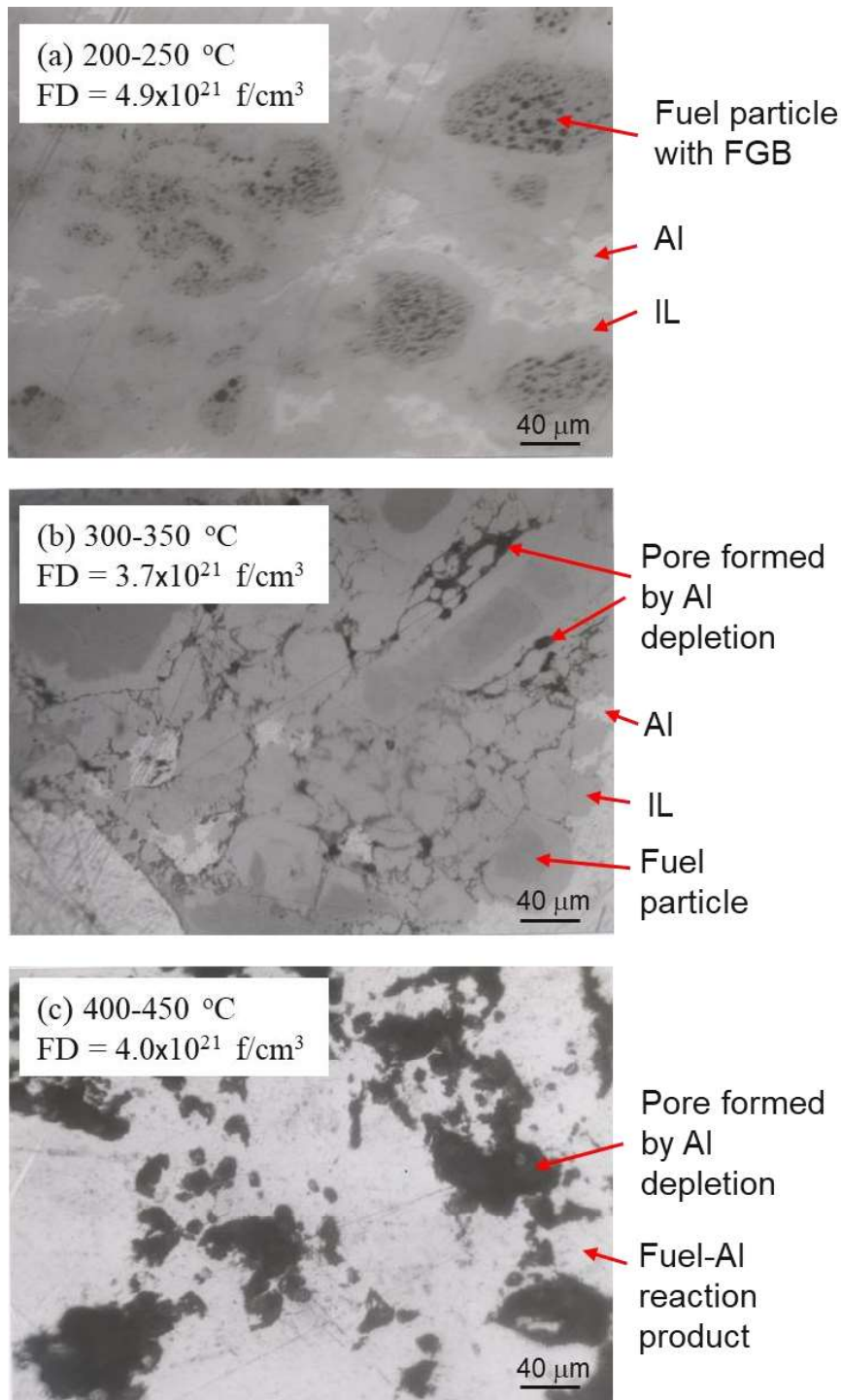


Figure 5: Low-enriched  $U_3Si_2$ -Al dispersion fuel samples with  $4.8 \text{ gU/cm}^3$  irradiated for 23 effective full power days (EFPD) at the indicated fuel temperatures. Temperatures were controlled during the irradiation, so they are not proportional to fission densities (FD) [7].

### 2.3 Effect of Uranium Density

When the U-density of  $U_3Si_2$ -Al dispersion fuel increases, the thermal conductivity and initial Al volume fraction of the fuel meat decrease. The majority of in-pile test data are for U-density

of 4.8 gU/cm<sup>3</sup>-fuel meat. Although limited, there are some data for U-densities higher than 4.8 gU/cm<sup>3</sup>. The test in the ORR [2] included miniplates with higher U-densities up to 5.7 gU/cm<sup>3</sup>. At low temperatures ( $\leq \sim 130$  °C), the increase in U-density exhibited no significant impacts on fuel performance [2].

Although there was minimal impact of increasing the U-density in the fuel at lower powers, at high powers the U-density can significantly impact the fuel performance. The increased surface area and decreased Al matrix increases the rate at which the matrix is consumed by IL formation, leading to large pores (as seen in Figure 5(c)). For U-density of 6.1 gU/cm<sup>3</sup> under high power [8], complete depletion of the Al matrix occurred at 25% burnup (FD =  $1.3 \times 10^{21}$  f/cm<sup>3</sup>, peak temperature reaching  $\sim 420$  °C), a much lower burnup than when depletion occurs at lower power (see Figure 5 (c)). Note that the test conditions of the test of Ref. [8] are different from those of the test shown in Figure 5 (c), making direct comparisons difficult. Establishing a U-density threshold for each power regime is desirable. A systematic set of data and a thorough evaluation of irradiation parameters are needed in order to compile this information.

### **3. U<sub>3</sub>Si<sub>2</sub>-Al PROPERTIES**

The properties of U<sub>3</sub>Si<sub>2</sub>, matrix Al, and the U<sub>3</sub>Si<sub>2</sub>-Al meat composite necessary for fuel designs and safety analyses were compiled in the report. Primarily, only un-irradiated material properties are available. Many of the material properties are considered to be insensitive to irradiation. However, thermal conductivity, density, and heat capacity may change due to irradiation, for which no data are currently available. These items are identified as gaps.

### **4. GAP ANALYSIS AND METHODS OF RESOLUTION**

The availability of fuel performance information and thermo-physical properties of U<sub>3</sub>Si<sub>2</sub>-Al under higher power conditions are summarized in Table 1. The proposed methods and plans to resolve the data gaps are also included. Note here that there are on-going irradiation tests of the U<sub>3</sub>Si<sub>2</sub>-Al system at those higher power conditions in BR-2 (e.g., HiPROSIT experiment [9]), which are expected to fill in some of the data gaps identified in the table. Only limited data from the EVITA testing are currently available, but included in the table in case more data become available in the future.

Table 1: Gap analysis and resolution methods

Topic	Gap reason	Method to fill gap
Specifications	<ul style="list-style-type: none"> <li>- No gap for currently qualified fuel, but modification for high power use may be needed.</li> <li>- Inspection methods to be developed with fabricators.</li> </ul>	<ul style="list-style-type: none"> <li>- Decide when new data are available</li> </ul>
Fuel meat swelling at high power conditions	<ul style="list-style-type: none"> <li>- Quantitative meat swelling data for high-power tests are needed.</li> <li>- FGB swelling at higher power is not quantitatively known.</li> </ul>	<ul style="list-style-type: none"> <li>- Existing test data from BR2 [8]</li> <li>- EVITA data</li> <li>- On-going Tests at BR2</li> </ul>
Interdiffusion reaction between $U_3Si_2$ and Al at high temperatures	<ul style="list-style-type: none"> <li>- Al matrix consumption kinetics is critical to sound fuel performance, but data are scarce at high temperatures.</li> </ul>	<ul style="list-style-type: none"> <li>- Existing test data from BR2 [8]</li> <li>- EVITA data</li> <li>- On-going Tests at BR2</li> </ul>
Effect of higher fuel loading at high power conditions	<ul style="list-style-type: none"> <li>- Performance data on the effect of U-loading higher than <math>4.8 \text{ gU/cm}^3</math> at high power (or high temperature) are needed.</li> <li>- A threshold function on fuel loading versus power may be desirable.</li> </ul>	<ul style="list-style-type: none"> <li>- Thorough re-analysis of existing data from legacy tests from ORR and BR2</li> <li>- On-going tests at BR2</li> </ul>
Fuel meat creep	<ul style="list-style-type: none"> <li>- The effect of this phenomenon is not fully understood.</li> <li>- Quantitative creep data are not available.</li> </ul>	<ul style="list-style-type: none"> <li>- Thorough re-analysis of existing data from legacy tests from ORR and BR2</li> <li>- EVITA data</li> <li>- On-going tests at BR2</li> </ul>
Effect of burnable absorber (BA) addition	<ul style="list-style-type: none"> <li>- Some studies are available for low-power tests, but more data are needed at high power conditions.</li> </ul>	<ul style="list-style-type: none"> <li>- On-going tests in BR2</li> </ul>
Un-irradiated $U_3Si_2$ thermal conductivity	<ul style="list-style-type: none"> <li>- Data for bulk <math>U_3Si_2</math> are tentatively used for <math>U_3Si_2</math> powder. Values for the powder may be different.</li> </ul>	<ul style="list-style-type: none"> <li>- Measurement of particulate <math>U_3Si_2</math></li> </ul>
Irradiated $U_3Si_2$ thermal conductivity	<ul style="list-style-type: none"> <li>- No measured data are available.</li> <li>- Modeling may be more time and cost effective.</li> </ul>	<ul style="list-style-type: none"> <li>- Low scale modelling</li> <li>- Measurement of irradiated fuel</li> </ul>
Density of irradiated fuel meat	<ul style="list-style-type: none"> <li>- Some measurement data are available.</li> <li>- More measurement data are needed.</li> <li>- Modeling may help.</li> </ul>	<ul style="list-style-type: none"> <li>- On-going tests in BR2</li> <li>- Model needs to be verified with measured data</li> </ul>
Heat capacity of irradiated fuel meat	<ul style="list-style-type: none"> <li>- Heat capacity of irradiated fuel meat changes by meat density change (e.g., FGB).</li> <li>- No measurement data are available.</li> <li>- Modeling may be a realistic option.</li> </ul>	<ul style="list-style-type: none"> <li>- Modelling may be needed</li> <li>- Measurement of irradiated fuel</li> </ul>
Properties of EU cladding materials	<ul style="list-style-type: none"> <li>- EU cladding (AG3NE or AlFeNi) data are less known than US test cladding AA 6061.</li> </ul>	<ul style="list-style-type: none"> <li>- Collaboration with CEA and SCK CEN</li> </ul>
Blister threshold temperature	<ul style="list-style-type: none"> <li>- Low power data exist.</li> <li>- Plates irradiated at high power may have different blister temperatures.</li> </ul>	<ul style="list-style-type: none"> <li>- Collaboration with CEA and SCK CEN</li> </ul>



## 5. SUMMARY

A report on U<sub>3</sub>Si<sub>2</sub>-Al dispersion fuel has been drafted. A brief summary of the report focusing on fuel performance and properties of the fuel is presented. The data gaps to establishing a technical basis of this fuel performance at higher power conditions are identified and the resolution methods of the gaps are proposed.

At low power conditions (having a heat flux  $\leq \sim 140$  W/cm<sup>2</sup> and a peak fuel temperature  $\leq \sim 130$  °C), U<sub>3</sub>Si<sub>2</sub> fuel performs well, showing low and stable fuel swelling up to 100% LEU burnup. FGBs were small and, more importantly, they were contained within the fuel particles. At the combination of higher power conditions and temperatures, while the kinetics of FGB swelling does not appear to change significantly, enhanced IL growth consumed the Al matrix more rapidly and, in some extreme cases, a complete depletion of matrix Al occurred. Al matrix depletion and subsequent pore formation in the meat could potentially lead to pillowing.

Increasing the U-density was not significantly disadvantageous to fuel performance under low power conditions. However, at high power conditions, the increased U-loading could have a more significant effect on fuel performance by increasing the potential for a more rapid depletion of the Al matrix and needs further investigation. The effects of the combination of high power and high temperature will be further clarified with the availability of the data from the on-going and planned irradiation tests.

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